

Machine-learned force fields

18-04-24. 27th Wien2k Workshop. Trieste.

Georg Madsen

Institute of Materials Chemistry, TU Wien, Austria



TECHNISCHE
UNIVERSITÄT
WIEN

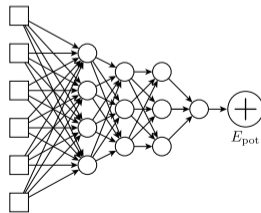
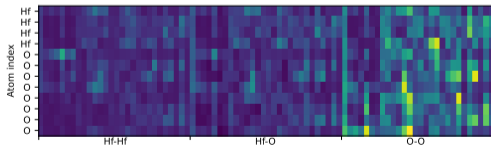
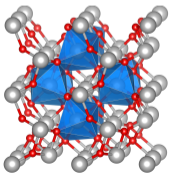
- Neural-network force fields
- Effective harmonic potentials
 - Phase stability in HfO_2
- The SrTiO_3 surface phase diagram
- Nested sampling
 - The phase diagram of Si

Machine-learned force fields

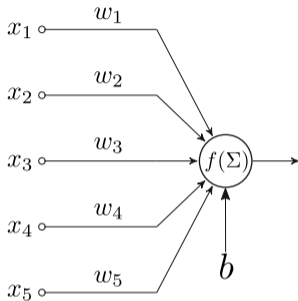
Use regression model to bypass explicit calculation of E_{pot}

Descriptor based MLFF:

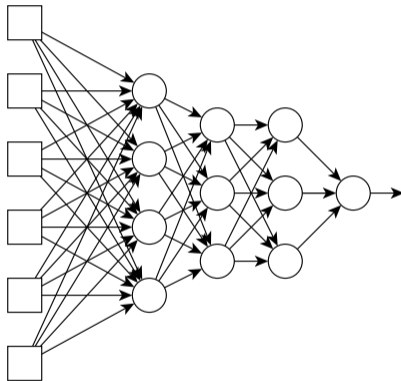
- Behler-Parinello (2007) and Bartók-Csányi (2010)
- Preprocess structure into symmetry compliant fingerprints
- Map onto energies and forces through ML model



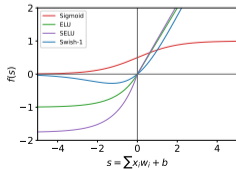
Perceptron



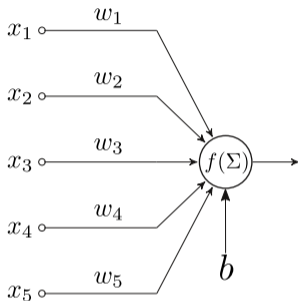
Multi-layer perceptron (Neural network)



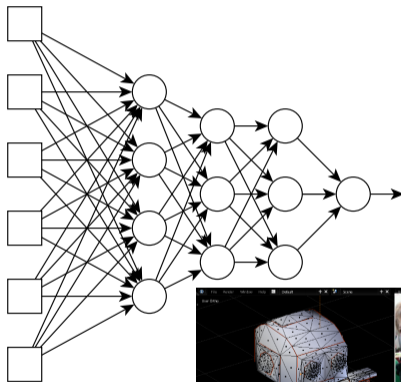
Activation function



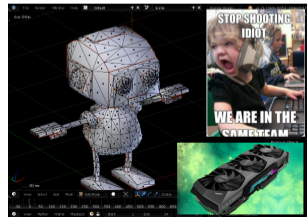
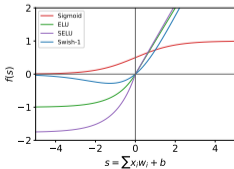
Perceptron



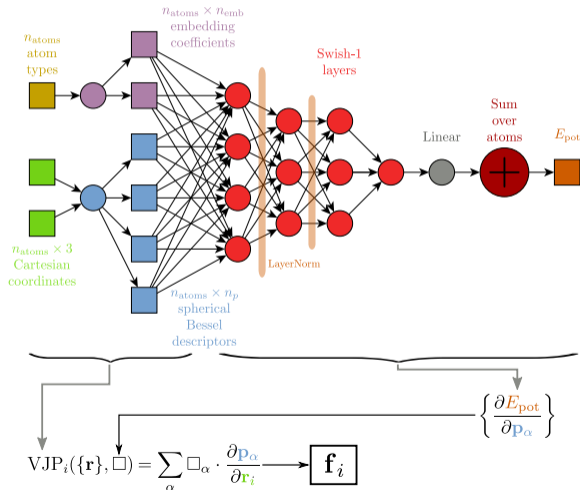
Multi-layer perceptron (Neural network)



Activation function



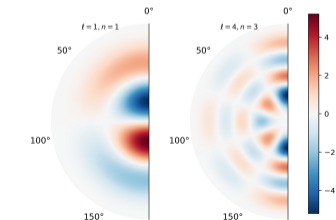
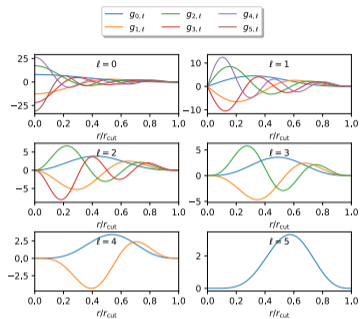
Algorithmically differentiable JAX-based neural-network force field



- Just-in-time compilation of Python functions
- Automatic differentiation
- Multiple CPU/GPU/TPU cores

- Swish activation function
- Element-specific-spherical-Bessel fingerprints

Spherical Bessel descriptors



Kocer, Mason, Erturk AIP Adv. 10 (2021) 015021

Project weighted local density

$$\rho_i(\mathbf{r}) = \sum w_{ij} \delta(\mathbf{r} - \mathbf{R}_{ij})$$

onto orthonormal basis functions

$$B_{nlm} = g_{n-l,l}(r) Y_l^m(\theta, \varphi)$$

and calculate the power spectrum.

Optimize Completeness

Training. 15 ionic pairs of ethylammonium nitrate

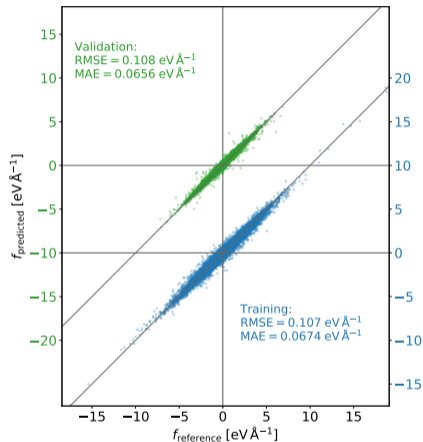
Project local density of each chemical species

$$\rho_{ij}(\mathbf{r}) = \sum \delta(\mathbf{r} - \mathbf{R}_{ij})$$

onto orthonormal basis functions

$$B_{nlm} = g_{n-l,l}(r)Y_l^m(\theta, \varphi)$$

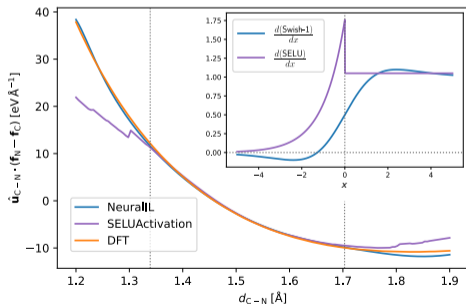
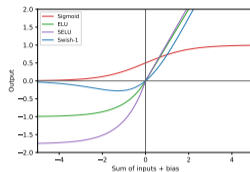
Model	MAE E_{pot} (meV/atom)	MAE f meV/Å
ZWEIGHTS	16.9	171
NEURALIL	1.9	65
NEURALIL+CENT	1.9	61



Swish activation function

$$s_{\beta}(x) = \frac{x}{1 + \exp(-\beta x)}$$

Ramachandran, Zoph, Le *arXiv 1710.05941*



Diffusion coefficients

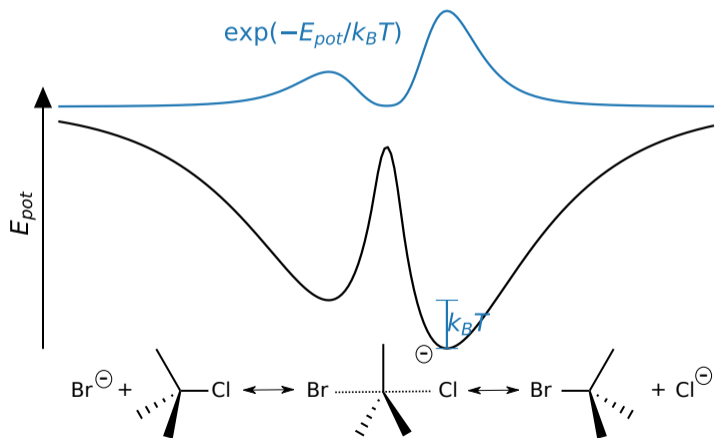
	$10^{-11} \text{ m}^2/\text{s}$	Exp	OPLS	NeuralIL
D_{anion}		6.9	0.13	8.7
D_{cation}		4.6	0.07	8.2

- Neural-network force fields
- Effective harmonic potentials
 - Phase stability in HfO_2
- The SrTiO_3 surface phase diagram
- Nested sampling
 - The phase diagram of Si

Canonical ensemble

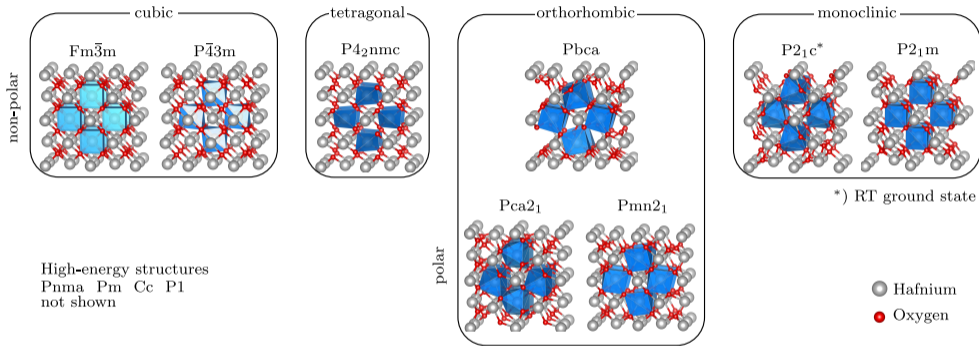
Partition function

$$Z(N, V, T) \propto \sum \exp\left(\frac{-E_{pot}}{k_B T}\right)$$



$$20^{3.6} = 2.6 \cdot 10^{23}$$

Hafnia



- Gate dielectrics semiconductor devices
- Control rods in nuclear reactors
- Ferroelectricity
- Surprisingly poor understanding of phase diagram

The Zirconia–Hafnia System: DTA Measurements and Thermodynamic Calculations

Chong Wang, Matvei Zinkevich[†], and Fritz Aldinger^{*}

Max-Planck-Institut für Metallforschung and Institut für Nichtmetallische Anorganische Materialien, Universität Stuttgart, Stuttgart, Germany

Table II. Literature Information on the Phase Transition Data of HfO₂ (α , Monoclinic; β , Tetragonal; γ , Cubic; *L*, Liquid)

Transition	Reference	Temperature (K)			Enthalpy of transformation (J/mol)	Entropy of transformation (J · (mol · K) ⁻¹)	Experimental method
		<i>A_s</i>	<i>M_s</i>	<i>T₀</i>			
$\alpha \leftrightarrow \beta$	Curtis <i>et al.</i> ²⁶	1973					HTXRD
	Wolten ⁶	1883	2013	1948			HTXRD
	Aldebert <i>et al.</i> ²⁴	1773	1823	1798			HTXRD
	Fujimori <i>et al.</i> ¹⁸	2113	2063	2088			Dilatometry
	Boganov <i>et al.</i> ²⁷	2173					HTXRD
	Ruh <i>et al.</i> ¹¹			1893			HTXRD
	Senft and Stubican ²²			2023 ± 20			HTXRD
	Stacy <i>et al.</i> ²³			2038			ND
	Stacy and Wilder ²⁵	2023	2073	2048			HTXRD
	Ruh and Hollenberg ²⁸	1863					HTXRD
	Stansfield ¹⁹	2073					DTA
	Gulamova and Novoselova ²¹		2066 ± 40				HTXRD
	Kasper and Troyanchuk ⁵			2023 ± 20			HTXRD
	Shevchenko <i>et al.</i> ¹⁴	2103	2083	2093			DTA
	Kuznetsov <i>et al.</i> ²⁰	2073					HTXRD
	Baun ¹⁷	2080 (<i>A_T</i>)	2018				RS
	This work	2066	2038	2052			DTA, extrapolated
Boganov <i>et al.</i> ²⁷	2973					HTXRD	
Kasper and Troyanchuk ⁵	2873					DTA	
Shevthenko and Lopato ²⁹	2803					DTA	

Effective harmonic potentials

$$F_{\text{harm}}[\{\mathbf{a}_i\}, T] = \sum_{n\mathbf{q}} \left(\frac{\hbar\omega_{n\mathbf{q}}}{2} + k_B T \ln \left[1 - e^{-\frac{\hbar\omega_{n\mathbf{q}}}{k_B T}} \right] \right)$$

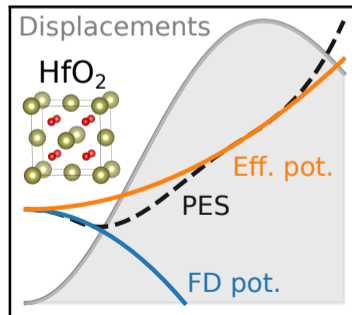
Determine the best harmonic approximation for the part of the potential energy surface which dominates nuclear motion at given conditions (Hooton, 1955)

Variational formulation

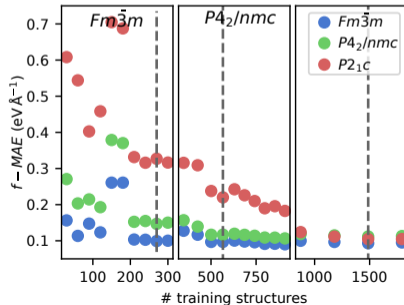
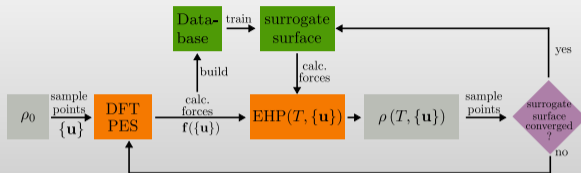
$$F[\hat{U}, \hat{\rho}_0] \leq \mathcal{F}_{\text{EHP}} = \underbrace{F[\hat{u}, \hat{\rho}]}_{F_{\text{harm}}} + \underbrace{\text{Tr}[\hat{\rho}(\hat{U} - \hat{u})]}_{F_{\text{corr}}}$$

$$F_{\text{corr}}[\{\mathbf{a}_i\}, T] = \int \rho(E_{\text{pot}} - E_{\text{HA}}) d\mathbf{R}$$

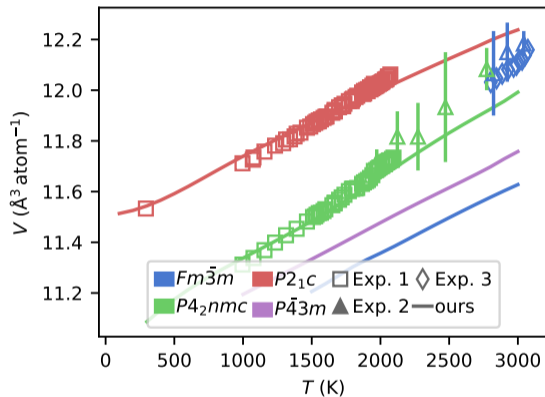
Errea, Calandra, Mauri *Phys. Rev. Lett.* 111 (2013) 177002



HfO₂: Effective harmonic potential with machine-learned force field

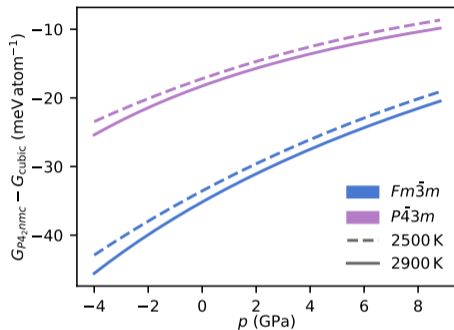
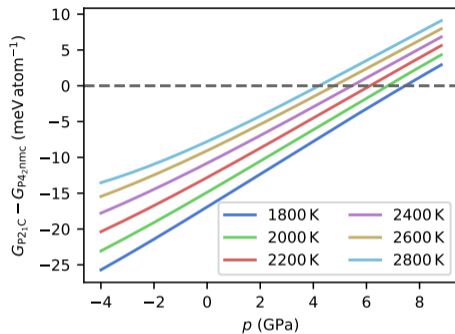


HfO₂ Thermal expansion



- Approx. $3 \cdot 10^6$ structures evaluated
- Excellent agreement of temperature dependence of lattice constants
- Not for cubic phases

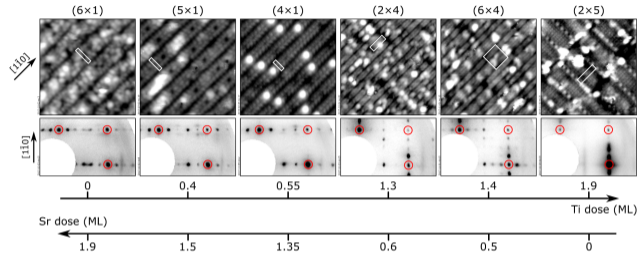
HfO₂ Free energies



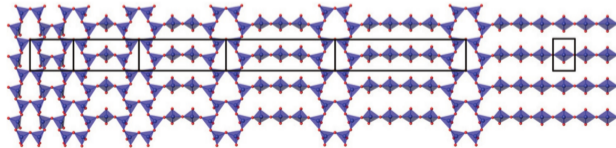
- Phase transition $P2_1c - P4_2nmc$
- Cubic phases not stable
 - Few experiments with internal disagreement
 - Presence of oxygen vacancies ?
 - Multiple results suggest that no stoichiometric cubic phase exists in ZrO₂

- Neural-network force fields
- Effective harmonic potentials
 - Phase stability in HfO_2
- The SrTiO_3 surface phase diagram
- Nested sampling
 - The phase diagram of Si

SrTiO₃(110) surface phase diagram



Riva et al., *Phys. Rev. Mater.* 3 (2019) 043802



Enterkin et al., *Nat. Mater.* 9 (2010) 245
see also Wang et al. *Phys. Rev. B* 90 (2014) 035436

The Covariance Matrix Adaptation - Evolution Strategy

- ESs commonly employed for continuous-parameter optimisation
- Employ lower variety of genetic operators than GA
- Endogenous parameters define the genotype distribution and are adapted during evolution



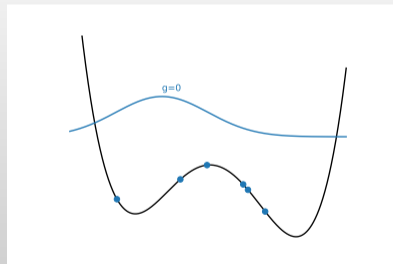
GitHub: Clinamen2

CMA-ES

- Population drawn from multi-dimensional normal distribution

$$\mathbf{x}_k^{(g)} \sim \mathcal{N}(\mathbf{m}^{(g)}, (\sigma^{(g)})^2 \mathbf{C}^{(g)})$$

- Transferable hyperparameters
- Problem-dependent parameters: $\sigma^{(0)}$, $\mathbf{m}^{(0)}$



The Covariance Matrix Adaptation - Evolution Strategy

- ESs commonly employed for continuous-parameter optimisation
- Employ lower variety of genetic operators than GA
- Endogenous parameters define the genotype distribution and are adapted during evolution



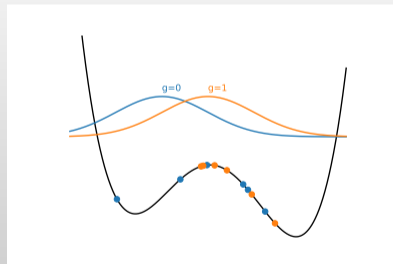
GitHub: Clinamen2

CMA-ES

- Population drawn from multi-dimensional normal distribution

$$\mathbf{x}_k^{(g)} \sim \mathcal{N}(\mathbf{m}^{(g)}, (\sigma^{(g)})^2 \mathbf{C}^{(g)})$$

- Transferable hyperparameters
- Problem-dependent parameters: $\sigma^{(0)}$, $\mathbf{m}^{(0)}$



The Covariance Matrix Adaptation - Evolution Strategy

- ESs commonly employed for continuous-parameter optimisation
- Employ lower variety of genetic operators than GA
- Endogenous parameters define the genotype distribution and are adapted during evolution



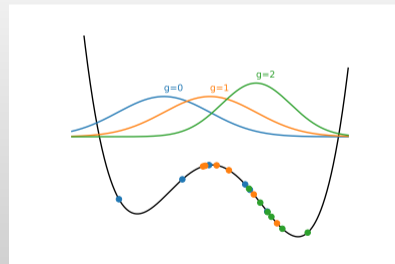
GitHub: Clinamen2

CMA-ES

- Population drawn from multi-dimensional normal distribution

$$\mathbf{x}_k^{(g)} \sim \mathcal{N}(\mathbf{m}^{(g)}, (\sigma^{(g)})^2 \mathbf{C}^{(g)})$$

- Transferable hyperparameters
- Problem-dependent parameters: $\sigma^{(0)}$, $\mathbf{m}^{(0)}$



The Covariance Matrix Adaptation - Evolution Strategy

- ESs commonly employed for continuous-parameter optimisation
- Employ lower variety of genetic operators than GA
- Endogenous parameters define the genotype distribution and are adapted during evolution



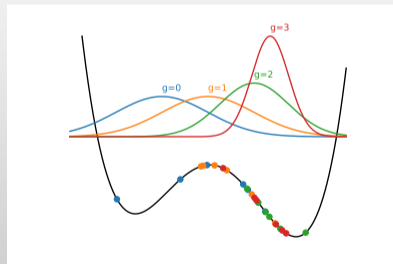
GitHub: Clinamen2

CMA-ES

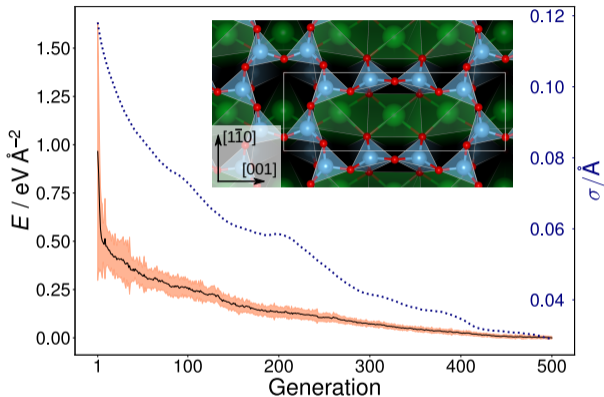
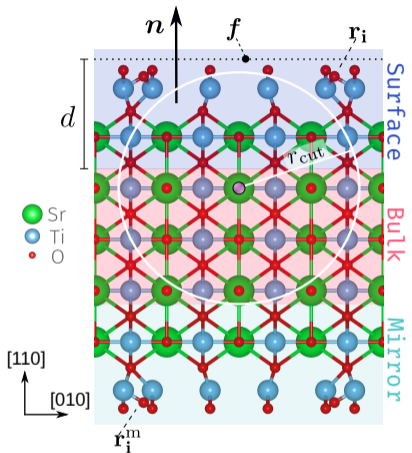
- Population drawn from multi-dimensional normal distribution

$$\mathbf{x}_k^{(g)} \sim \mathcal{N}(\mathbf{m}^{(g)}, (\sigma^{(g)})^2 \mathbf{C}^{(g)})$$

- Transferable hyperparameters
- Problem-dependent parameters: $\sigma^{(0)}$, $\mathbf{m}^{(0)}$

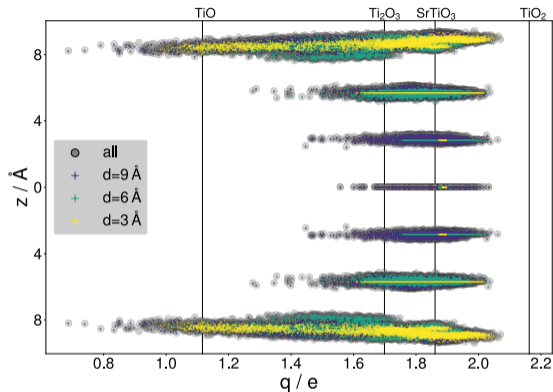
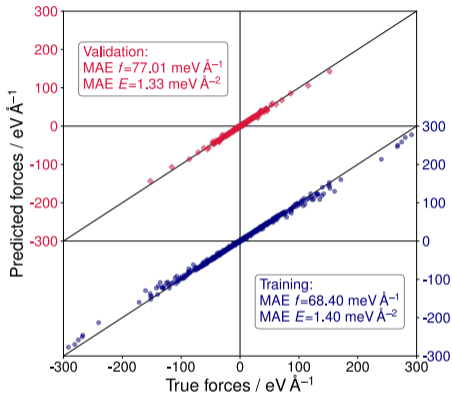


Surface reconstructions with CMA-ES



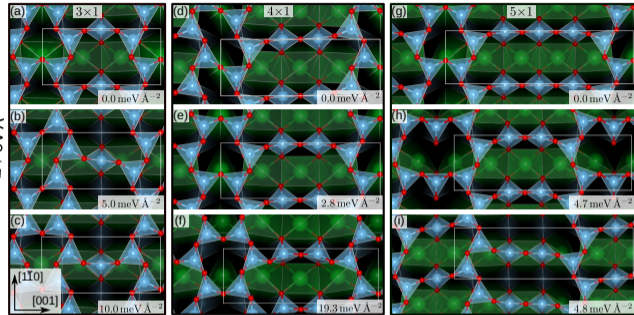
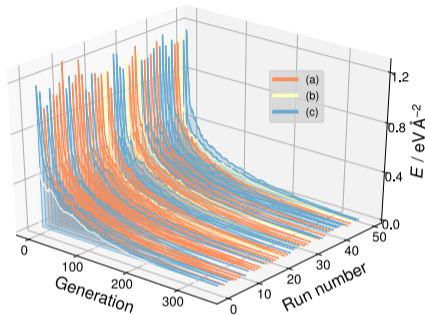
- Adapt CMA-ES to surface reconstructions
- 42 atoms \rightarrow 126 dof (4×1)
- DFT backend reproduces 6-10 overlayer from literature

Training NNFF on CMA-ES trajectory



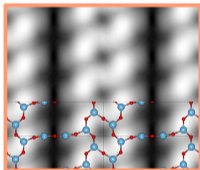
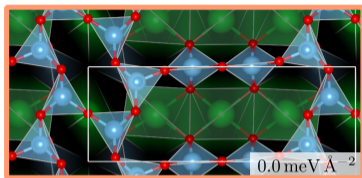
- Good agreement with DFT over large range of energies/forces
- Diverse set of training structures with net atomic charges covering known titanium oxidation states.

Sets of CMA-ES runs with NNFF backend

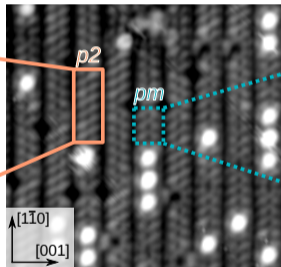


- Set of 50 runs. Same founder; Different random seeds
- 4×1 : $P2$ structure energetically comparable to Pm structure
- 4×1 training data.

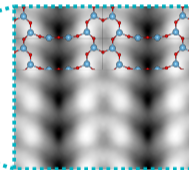
STO(110) 4 × 1 - STM



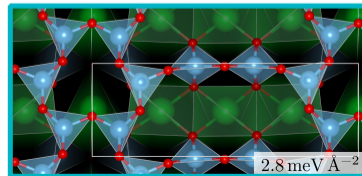
simulated



experimental¹



simulated



Wanzenböck et al., *Digit. Discov.* 1 (2022) 703

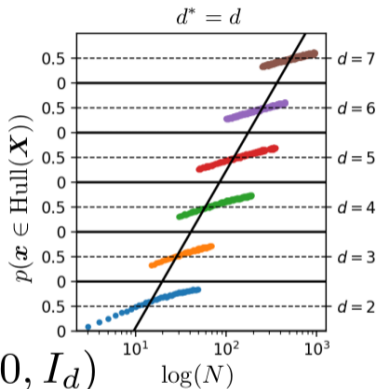
¹ Experimental: Wang et al., *Nano Lett.* 16 (2016) 4

Learning in High Dimension Always Amounts to Extrapolation

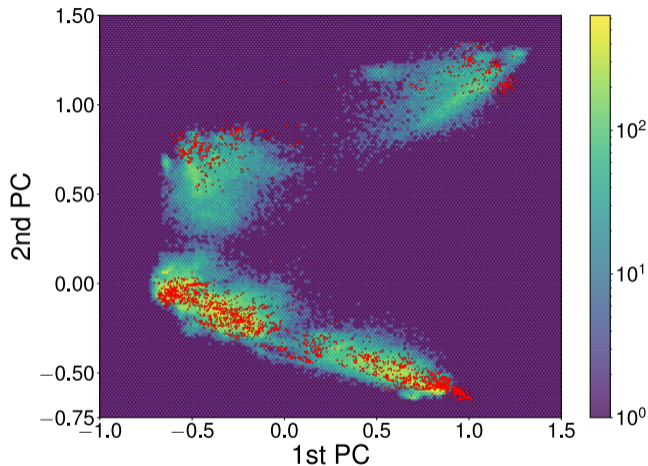
Randall Balestriero¹, Jérôme Pesenti¹, and Yann LeCun^{1,2}¹Facebook AI Research, ²NYU

The notion of interpolation and extrapolation is fundamental in various fields from deep learning to function approximation. Interpolation occurs for a sample \mathbf{x} whenever this sample falls inside or on the boundary of the given dataset's convex hull. Extrapolation occurs when \mathbf{x} falls outside of that convex hull. One fundamental (mis)conception is that state-of-the-art algorithms work so well because of their ability to correctly interpolate training data. A second (mis)conception is that interpolation happens throughout tasks and datasets, in fact, many intuitions and theories rely on that assumption. We empirically and theoretically argue against those two points and demonstrate that on any high-dimensional (>100) dataset, interpolation almost surely never happens. Those results challenge the validity of our current interpolation/extrapolation definition as an indicator of generalization performances.

$$\mathbf{x}_i \sim \mathcal{N}(0, I_d)$$

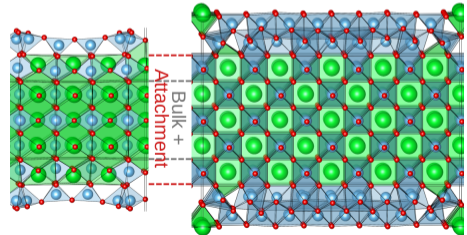
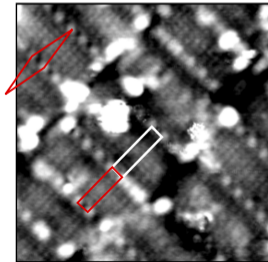
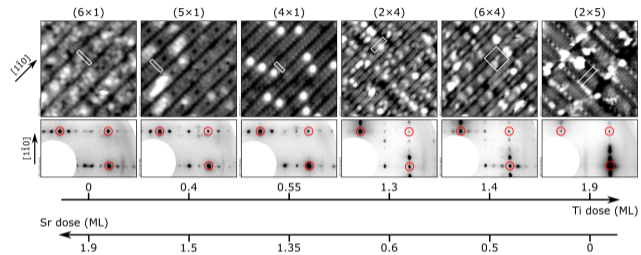


PCA fingerprints.



Similarity: The projection of the local atomic environments of the low energy 5×1 structures fall within the area covered by the projection of the 4×1 training data.

STO(110) $2 \times m$

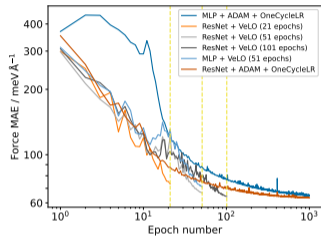


Uncertainty estimation

- Deep ensembles
- Committees

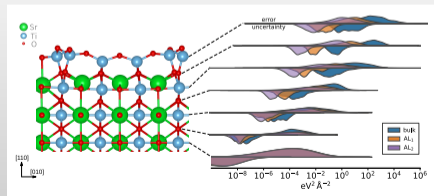


Versatile Learned Optimization

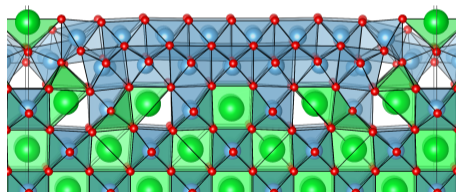
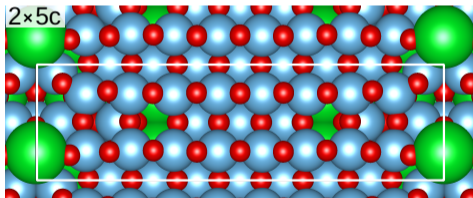
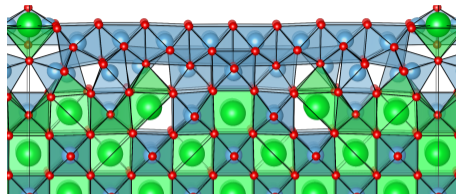
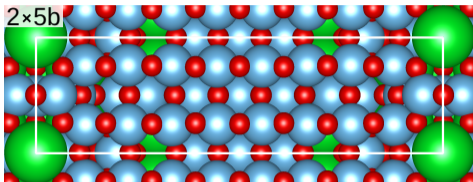


Optimize adversarial loss
(Koda-Bombarelli)

$$\mathcal{L} = \sigma_f^2 \exp\left(-\frac{E_{\text{pot}}}{k_B T}\right)$$



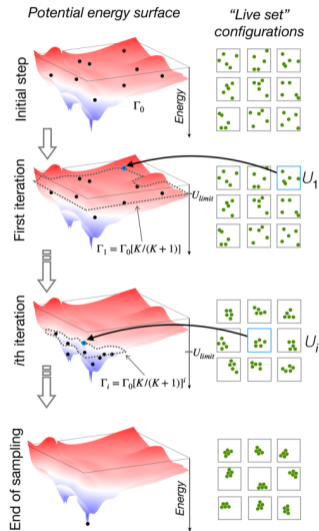
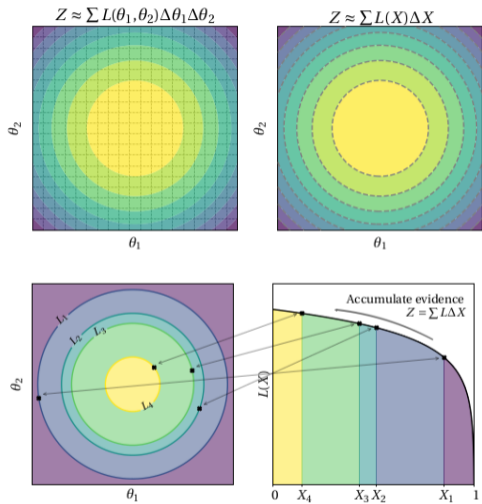
STO(110) $2 \times m$: MLFF driven EA



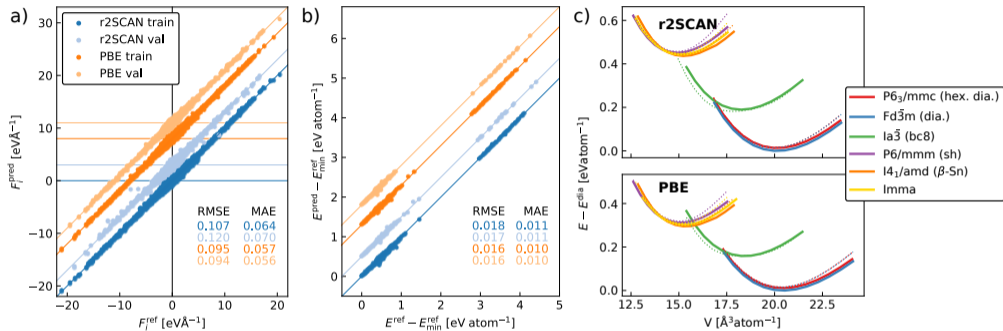
- 158 atoms: 474 DOF
- Identify several local minima
- New structure in agreement with published STM

- Neural-network force fields
- Effective harmonic potentials
 - Phase stability in HfO_2
- The SrTiO_3 surface phase diagram
- Nested sampling
 - The phase diagram of Si

Skilling's nested sampling

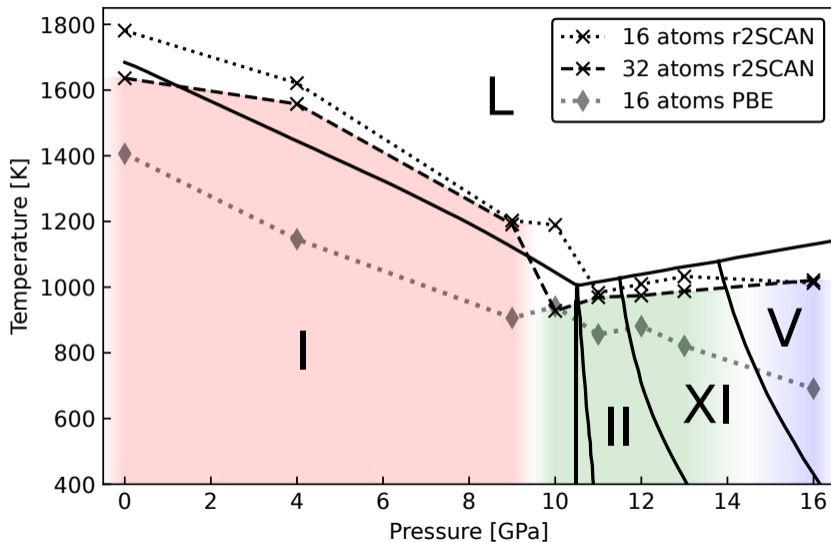


Silicon. The NNFF

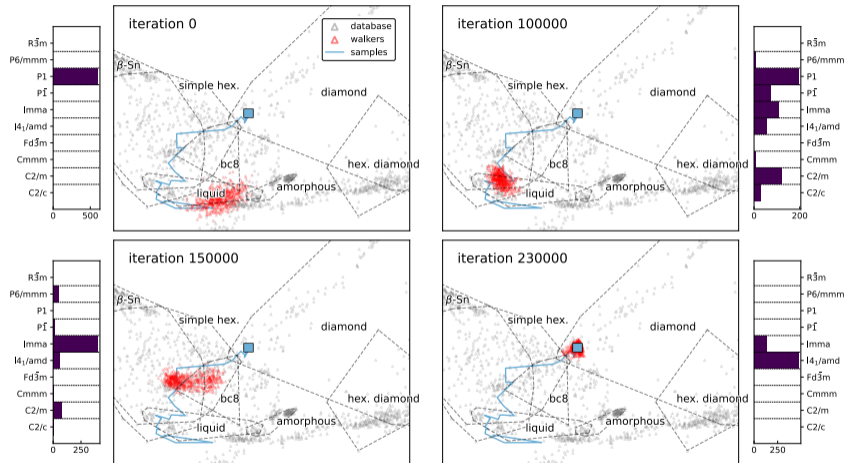


- Structures taken from database of Bartók-Csányi and recalculated with PBE and r2SCAN
- r2SCAN results in increase in energy differences

Silicon. The phase diagram

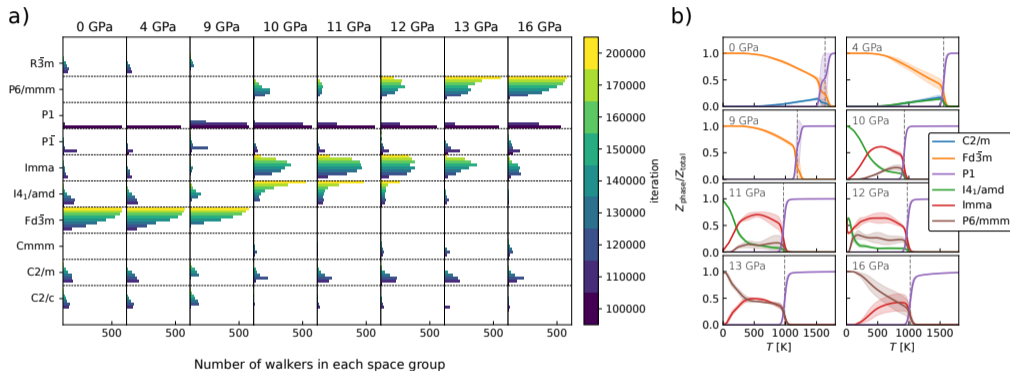


Silicon. The nested sampling walkers



- Highly diverse training set is essential

Silicon. The basins



Unglert, Carrete, Pártay, Madsen. *Phys. Rev. Mater.* 7, (2023) 123804